Morphological, Structural and Optical Properties Study of Transition Metal Ions Doped TiO₂ Nanotubes Prepared by Hydrothermal Method

Mohd Hasmizam Razali, Ahmad-Fauzi M. N., Abdul Rahman Mohamed, and Srimala Sreekantan

Abstract-Undoped TiO₂ nanotubes and doped titanium dioxide nanotubes with various transition metal ions (Co²⁺, Ni²⁺, Cu²⁺) were prepared by hydrothermal method. The samples characterized using X-ray were diffraction (XRD). field-emission scanning electron microscopy (FESEM), transmission electron microscopy (TEM), nitrogen gas adsorption and UV-Vis diffuse reflectance spectroscopy. The structural and morphological studies showed that the transition metal ion dopant was incorporated into interstitial positions of the TiO₂ lattice to form a new phase of TiO₂ (hexagonal) instead of anatase TiO₂ (tetragonal) for undoped TiO₂ nanotubes. It could be expected that doped TiO₂ nanotubes active under visible light due to their small band gap.

Index Terms—Doped, photocatalyst, TiO_2 nanotubes, transition metal ion.

I. INTRODUCTION

Recently, extensive researches have been conducted on the synthesis and characterization of TiO2 nanotubes because of their novel properties such as unique shape, size confinement in radial-direction, large specific surface area and large pores volumes [1]. They also have high interfacial charge transfer rate, resulting in the carriers being free to move throughout the length of these nanostructures thus, reduce the e^{-}/h^{+} recombination probability [2]. However, due to the their large band gap energy (~ 3.25 eV) [3], it only active only under UV-light spectral range, which is a small fraction (5-6) % of whole solar-light spectrum [4], [5]. Therefore, doped titanium dioxide nanotubes photocatalyst have been extensively investigated due to their capability to extend the wavelength response of TiO2 nanotubes into the visible region and inhibits phase transformation. Moreover, doping enhance its photoactivity by reducing could the recombination of e⁻ and h⁺ pair and enhancing the interfacial

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Abdul Rahman Mohamed is with the School of Chemical Engineering, Universit Sains Malaysia, Engineering Campus, Seri Ampangan, 14300 USM, Nibong Tebal, Pulau Pinang, Malaysia (e-mail: chrahman@eng.usm.my) charge transfer [6], [7]. So far, various transition metal ions [8], [9], noble metal [10], and non-metals [11]-[13] doped TiO₂ have been successfully prepared and investigated. However, there is a considerable controversy on the effect doping metal-ions on photocatalytic performance. Some papers reported that the doping of metal-ions such as W^{6+} and Cu^{2+} in TiO₂ increased the photocatalytic performance [14], [15], whereas Nagaveni et al., demonstrated that TiO₂ doping with these metal-ions through solution combustion method showed weaker photocatalytic performance than pure TiO₂ [16]. Such wide variations in reported literatures should be mainly due to different preparation methods and doping elements which resulted in different physical properties of TiO₂. Thus, in this present work, various transition metal ions doped TiO₂ nanotubes were synthesized by hydrothermal method and were characterized using FESEM, TEM, XRD, nitrogen gas sorption and UV-Vis diffuse reflectance spectroscopy.

II. EXPERIMENTAL

A. Preparation of Undoped and Doped TiO₂ Nanotubes

To synthesis undoped TiO₂ nanotube, 2 g of the commercial TiO₂ powder precursor (merck) was mixed with 100 mL of 10 M NaOH. The mixture was subjected to hydrothermal treatment at 150 °C for 24 hours in autoclave. When the reaction was completed, the white solid was collected and washed with 0.1 M HCl (200 ml) followed by distilled water until a pH 7 of washing solution was obtained. The final products were obtained by the filtration with subsequent drying at 80 °C for 24 hours. Then, sample was heated at 300 °C for 2 hours in the air. In order to synthesis Co²⁺, Ni²⁺ and Cu²⁺ doped nanotubes, similar procedure was repeated with 2 g of TiO₂ precursor was mixed with 100 mL of 10 M NaOH consists of 0.25 mmol of Co(NO₃)₂, Ni(NO₃)₂ and Cu(NO₃)₂ aqueous solution as metal ions source respectively.

B. Characterization

ZEISS SUPRATM 35VP field emission scanning electron microscope (FESEM) and Philips CM12 transmission electron microscope (TEM) was used to investigate the morphology of the sample. X-Ray powder diffraction (XRD) analysis was performed using a Bruker D8 Diffractometer with Cu-K α ($\lambda = 1.54021$ Å) and scans were performed in step of 0.2°/second over the range of 2 θ from 10 up to 80°. Quantachrome ASiQwin - nitrogen gas adsorption was used to determine the surface area and UV-Vis was carried out

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using Perkin Elmer Lambda UV-Vis spectrometer for band gap measurements.

III. RESULT AND DISCUSSION

The XRD patterns of undoped TiO₂ nanotubes and doped TiO₂ nanotubes with different metal ion shown in Fig. 1. TiO₂ nanotubes revealed the presence of the peaks at 25.25 °, $37.52 \circ$, $48.02 \circ$, $53.58 \circ$, $54.88 \circ$, $62.61 \circ$, and $75.07 \circ$ which are assigned to TiO₂ anatase. Meanwhile for Co²⁺, Ni²⁺, Cu²⁺ doped TiO₂ nanotubes at 0.25 mmol, only three peaks at about 19.89 °, $24.57 \circ$ and $48.30 \circ$ was observed. These peaks were due to the hexagonal phase of TiO₂.



Fig. 1. XRD patterns of (a) undoped TiO_2 nanotubes and 0.25 mmol doped TiO_2 nanotubes with (b) Co^{2+} (c) Ni^{2+} (d) Cu^{2+} metal ion

The lattice parameters of TiO_2 nanotubes and various transition metal ion doped TiO_2 nanotubes based on the XRD patterns are listed in Table I.

Sample	<i>a</i> (A)	<i>c</i> (A)	Phase content (%)
TiO ₂ nanotubes	3.781	9.509	100 anatase TiO ₂
Co ²⁺ doped	5.05	6.51	100 TiO ₂ hexagonal
TiO ₂ nanotubes			
Ni ²⁺ doped	5.11	6.45	100 TiO ₂ hexagonal
TiO ₂ nanotubes			
Cu ²⁺ doped	5.12	6.38	100 TiO ₂ hexagonal
TiO ₂ nanotubes			
Anatase TiO ₂	3.784	9.519	-
(tetragonal)			
TiO ₂ hexagonal	5.29	6.13	-

TABLE I: LATTICE PARAMETERS AND PHASE CONTENT OF THE SAMPLES

It can be seen that TiO₂ nanotubes have lattice parameters (a- and c-axis) of 3.781 Å and 9.509 Å, respectively, in the unit cell based on the tetragonal Bravais lattice. In comparison with doped titania nanotubes, there exists derivations of *a* and *c* values. This derivation as well as formation of new phase after metal ion doping is most probably due to the incorporation of metal ion into interstitial positions of the TiO₂ lattice. The interstitial diffusion of metal ion into the TiO₂ lattice can strongly modify the lattice parameters. In addition, no additional peaks corresponding to the dopants were observed proving those dopants ions are successfully incorporated into the lattice site of TiO₂. However, researchers [17], [18] reported that no peaks of dopant were observed in XRD pattern due to low concentration of dopant loading into TiO₂ nanotubes. The

XRD cannot identify the dopant peaks because of their detection limit.

In order to study the effect of transition metal ion doping on morphology of nanotubes, the FESEM and TEM was carried out. Fig. 2 represents the FESEM micrographs of undoped TiO₂ nanotubes and doped TiO₂ nanotubes with different metal ions (Co^{2+} , Ni^{2+} , Cu^{2+}). Fibrous-like structures is shown by FESEM image for undoped TiO₂ nanotubes (Fig. 2a). The diameter of the fibrous is about 12 nm. On the other hand, after doping with different metal ion of Ni²⁺ (Fig. 2b), Co²⁺ (Fig. 2c) and Cu²⁺ (Fig. 2d), similar morphological characteristics was observed. This indicating that, the metal ion doping had little effect on fibrous-like structure may be due to small amount of metal ion loading.







Fig. 3. TEM micrographs of (a) undoped $\rm TiO_2$ nanotubes and 0.25 mmol doped $\rm TiO_2$ nanotubes with (b) $\rm Co^{2+}$ (c) $\rm Ni^{2+}$ (d) $\rm Cu^{2+}$ metal ion

Meanwhile, TEM micrographs of the undoped and doped TiO_2 nanotubes are shown in Fig. 3. Fig. 3a shows TEM images of the undoped TiO_2 nanotubes. The existence of hollow inside the fibrous-like structures indicated for nanotubes formation. The inner and outer diameters of the nanotubes are 4 nm and 12 nm respectively. After doping with metal ion, there were no obvious changes in their morphological structure. All doped TiO_2 nanotubes also showed the existence of hollow inside the fibrous-like structure indicating that nanotubes structures were retained (Fig. 3b, 3c, 3d). The nanotubes structures were connected

randomly. This will promotes the diffusion of reactants and products, thus enhancing the photocatalytic activity by facilitating access to the reactive sites on the surface of the photocatalyst [19]. In addition, photocatalytic reactions are believed to take place on the illuminated surface. The tube-like structure increases the surface areas significantly and, consequently, helps rapid mass transfer of the adsorbed molecules from bulk solution onto the catalyst surface, causing the photocatalytic process to be accelerated [20].

TABLE II: BET SURFACE AREA OF TIO2 PRECURSOR, UNDOPED TIO2 NANOTUBES AND DOPED TIO2 NANOTUBES

Samples	BET surface area (m²/g)
TiO ₂ precursor	9.280
Undoped TiO ₂ nanotubes	226.521
Co ²⁺ doped	
TiO ₂ nanotubes	240.441
Ni ²⁺ doped	
TiO ₂ nanotubes	242.262
Cu ²⁺ doped	
TiO ₂ nanotubes	245.075

As shown in Table II the BET surface area of the TiO₂ precursor is $9.280 \text{m}^2/\text{g}$. However for undoped TiO₂ nanotubes, the value increases drastically to 226.521 m^2/g . It is predicted that the inner and outer surfaces of tubes structure are the major reason for the increase in surface area, which was one of the new attribute of the TiO_2 nanotubes compared with TiO₂ precursor. On the other hand, the BET surface areas of doped TiO2 nanotubes are larger than undoped TiO₂ nanotubes due to the presence of dopant, which can reduced TiO₂ crystallization and grain growth during the heat decomposition step, thus producing larger surface area material [21]. For doped material, the trends show that the BET surface area increases from 240. 441 m^2/g to 241.262 m²/g and 245.075 m²/g upon the addition of transition metal ion series from left to right of the periodic table (Co^{2+} , Ni^{2+} , Cu^{2+}). The differential of transition metal ion size may be contributed to the deviation of surface area.



Fig. 4-7 shows the N_2 adsorption-desorption isotherms of undoped TiO₂ nanotube and doped TiO₂ nanotubes. The isotherm for all samples exhibits a typical IV-like isotherm with H1 hysteresis according to IUPAC classification [22]. The shapes of hysteresis loops have often been identified

with specific pore structures. Type H1 hysteresis appearing in the multilayer range of physisorption isotherms is usually associated with capillary condensation in mesopore structures. Mesoporous structures allowed rapid diffusion of various reactants and product molecules passing through the pores during photodegradation reaction [22].



Fig. 5. Isotherm plot of 0.25 mmol Co^{2+} doped TiO_2 nanotubes



Fig. 6. Isotherm plot of 0.25 mmol Ni²⁺ doped TiO₂ nanotubes



Fig. 7. Isotherm plot of 0.025 mmol Co2+ doped TiO2 nanotubes

The band gap energy of the undoped titania nanotubes and doped nanotubes (copper ion) were determined from the reflectance data obtained in reflectance spectra [Fig. 8 and 10]. A graph of [In $(R_{max}-R_{min}/R-R_{min})$]² versus photon energy

(band gap energy) was plotted, where R_{max} represents the maximum reflectance value in desired wavelength and R_{min} represents the minimum reflectance value in desired wavelength. The energy band gap was determined based on the intersection from the extrapolation of the straight line of the curve to the y-axis = 0 [23]. In this study, the band gap energy of the undoped TiO₂ nanotubes is about ~3.27 eV (Fig. 9). The doped titanium dioxide nanotubes (with copper ion) on the other hand, show very small in their band gap energy (~2.06 eV) (Fig. 11), and shifted to the blue region wavelength in reflectance spectra. This probably due to the formation of a new phase (hexagonal) of TiO₂



Fig. 8. Reflectance spectra of undoped TiO₂ nanotubes



Fig 9. Energy band gap of undoped TiO₂ nanotubes







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